

## COMPUTER MODELING OF MAGNETIC PROPERTIES OF DISORDERED BCC PHASE OF $\text{Fe}_{100-x}\text{Me}_x$ ALLOYS ( $\text{Me} = \text{Al, Ga, Ge}$ )

**M.A. Zagrebin<sup>1</sup>, I.A. Markovich<sup>2</sup>, A.S. Kuznetsov<sup>1</sup>, M.V. Matyunina<sup>1</sup>, A.V. Butakov<sup>1</sup>**

<sup>1</sup> Chelyabinsk State University, Chelyabinsk, Russian Federation

<sup>2</sup> South Ural State University, Chelyabinsk, Russian Federation

E-mail: miczag@mail.ru

**Abstract.** In this paper, within the framework of density functional theory for the disordered phase A2 of Fe-Me alloys ( $\text{Me} = \text{Al, Ga, Ge}$ ) the magnetic exchange constants  $J_{ij}$  are calculated and the Curie temperatures  $T_C$  are estimated in dependence of the concentration of  $\text{Me} = \text{Al, Ga, Ge}$  (in the range of  $0 \leq x \leq 14$  at. %) and the crystal lattice parameter (in the range of  $2,7 \leq a \leq 3,0$  Å). The  $J_{ij}$  distribution obtained in the paper demonstrates non-linear behavior. The highest ferromagnetic interaction in the first coordination sphere is  $\approx 23$  meV. In the third coordination sphere, antiferromagnetic exchange interaction is observed, which reaches a value of  $\approx -2,5$  meV. Magnetic transition temperatures calculated using the mean field approximation based on the experimental values and the  $a_0$  parameters calculated within the density functional theory show the presence of a concentration range of  $4 \leq x \leq 8$  at. % in which  $T_C$  increases. The cross-section of the calculated Curie temperature distribution shows that for all studied Fe-Me alloys with the lattice parameter  $a_0 = 2,94$  Å, it is possible to construct a dependence  $T_C(x)$  that is in quality agreement with the experimental one.

**Keywords:** *Fe-based alloys; ab initio calculations; magnetic exchange interaction; Curie temperature.*

### Introduction

The scientific community's interest in the research and development of multifunctional iron-based materials is constantly growing. The demand for environmentally friendly materials makes these alloys especially attractive. The unique combination of mechanical, magnetic, and electrical characteristics of Fe-Me alloys ( $\text{Me} = \text{Al, Ga, Ge}$ ) with high magnetostriction values in low magnetic fields [1, 2] is of particular interest for researchers and makes such alloys promising for the creation of sensors and actuators and other magnetomechanical devices. The ability to convert magnetic energy into mechanical energy and vice versa allows these alloys to be used in industry, power engineering, automotive engineering, and other areas. Thus, studies of Fe-Me alloys ( $\text{Me} = \text{Al, Ga, and Ge}$ ) remain relevant both from the point of view of fundamental science and practical application. This is confirmed by the constant growth in the number of publications in this field.

Experimental studies show that addition of nonmagnetic elements such as aluminum, gallium, or germanium into  $\alpha$ -Fe with a body-centered cubic (bcc) lattice allows for a significant increase in the magnitude of tetragonal magnetostriction in the resulting alloys compared to pure iron [2]. The phase diagrams of the  $\text{Fe}_{100-x}\text{Me}_x$  systems ( $\text{Me} = \text{Al, Ga, Ge}$ ) in the concentration range from 0 to 10 at. % reveal the presence of a disordered bcc structure A2 (symmetry group no. 229,  $Im\text{-}3m$ , with  $\alpha$ -Fe as prototype). In this case, similar behavior of parameters such as tetragonal magnetostriction and the Curie temperature is observed. In the single-phase A2 region, a gradual decrease in Curie temperature  $T_C$  is observed, which becomes sharper during the formation of the  $D0_3$  structure. This is accompanied by the appearance of a two-phase region, which affects the magnetic characteristics of the material [3].

The theoretical calculation of the transition temperature from ferromagnetic (FM) to paramagnetic (PM) state, presented in [4–7], showed an interesting feature of the obtained results. A complex approach using *ab initio* calculations and Monte Carlo modeling reveals a decreasing behavior of the  $T_C(x)$

curve dependences in the  $\text{Fe}_{100-x}\text{Al}_x$ ,  $\text{Fe}_{100-x}\text{Ga}_x$ , and  $\text{Fe}_{100-x}\text{Ge}_x$  alloys only in the concentration range of  $x > 12$  at. %. However, in the region of lower concentrations, the behavior of the curve cannot be reproduced, since studies show the opposite trend: the Curie temperature increases with a change in the concentration of  $Me = \text{Al}, \text{Ga}, \text{Ge}$  from zero to 12 at. %. The authors explain this behavior by an increase in the exchange interaction  $J_{ij}$  between iron atoms, as well as an increase in the magnetic moment of Fe atoms  $\mu_{\text{Fe}}$ . It is important to note the observed correlation between the values of  $J_{ij}$  and  $\mu_{\text{Fe}}$  and the lattice parameter: the larger the lattice parameter, the stronger the exchange interaction and the greater the magnetic moment per iron atom [5, 6]. Thus, the obtained results indicate a complex relationship between the structure, composition, and magnetic properties of  $\text{Fe}_{100-x}Me_x$  alloys ( $Me = \text{Al}, \text{Ga}, \text{Ge}$ ), which opens up new prospects for basic research into the mechanisms of forming magnetic properties in these materials.

In this paper, we analyze the influence of the crystal lattice parameter and the concentration of nonmagnetic atoms of  $Me = \text{Al}, \text{Ga}, \text{Ge}$  on the exchange interaction, as well as the Curie temperature of the A2 structure of  $\text{Fe}_{100-x}Me_x$  alloys ( $0 \leq x \leq 14$  at. %) using *ab initio* simulation.

## 1. Calculation details

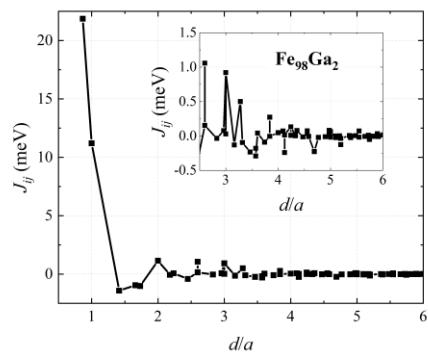
To simulate the magnetic properties in the  $\text{Fe}_{100-x}Me_x$  ( $Me = \text{Al}, \text{Ga}, \text{Ge}, 0 \leq x \leq 14$  at. %) system, we used the Korringa–Kohn–Rostoker Green's function method implemented in the SPRKKR software package (a Spin Polarized Relativistic Korringa–Kohn–Rostoker code) [8]. The fully disordered structure A2 was investigated in the work. For this phase, Fe and Me atoms are randomly distributed in the Wyckoff position  $2a(0, 0, 0)$ . To create non-stoichiometric compositions, we used the single-site coherent potential approximation (CPA). The use of CPA allows us to construct an averaged potential for a lattice site occupied by several types of atoms [8]. Note that this approximation successfully describes the properties of many compositionally disordered (non-stoichiometric) alloys. The crystal lattice parameter in the calculations was varied in the range of  $2,7 \leq a \leq 3,0 \text{ \AA}$ .

In the first stage, we performed calculations of the Heisenberg magnetic exchange constants  $J_{ij}$  in the formulation of A. Lichtenstein *et al.* [9]. The calculations included the use of the spin-polarized scalar-relativistic Dirac Hamiltonian with  $l_{\max} = 3$ . To perform both self-consistent and  $J_{ij}$  calculations, 4 495  $k$ -points were generated by a  $57 \times 57 \times 57$   $k$ -grid. For the exchange-correlation potential, the generalized gradient approximation in the Perdew–Burke–Ernzerhof (PBE) formulation was used [10]. Obtained exchange coupling constants were used to estimate the Curie temperature  $T_C$  in the mean field approximation [11].  $J_{ij}$  parameters in Curie, temperature estimation were taken into account up to the 49th coordination sphere and were limited to a value multiple of 6 lattice parameters.

## 2. Results and discussion

At the first stage, the magnetic exchange interaction parameters were calculated for the A2 structure of the  $\text{Fe}_{98}\text{Ga}_2$  alloy with the experimental lattice value  $a_0 = 2,87 \text{ \AA}$  [12]. Fig. 1 shows the values of  $J_{ij}$  between Fe atoms depending on the distance between the atoms (in units of the crystal lattice parameter). The exchange interaction between pairs of Ga-Ga and Fe-Ga atoms does not exceed 0,1 meV and is excluded from consideration. In Fig. 1,  $J_{ij}$  exhibit oscillating damped behavior. The largest FM interaction is observed in the first coordination sphere of Fe atoms and is  $\approx 22$  meV. Let us consider how the values of  $J_{ij}$  change depending on the crystal lattice parameter and the concentration of  $Me = \text{Al}, \text{Ga}$ , and Ge atoms.

Fig. 2 shows the distribution of the magnetic exchange interaction constants for the  $\text{Fe}_{100-x}\text{Al}_x$  alloy series depending on the value of the crystal lattice and the concentration of Al atoms for the first three coordination spheres (Fig. 2(a) – at a distance of  $\sqrt{3}/2a$ , Fig. 2(b) – at a distance of  $a$ , and Fig. 2(c) – at a distance of  $\sqrt{2}a$ ). The  $J_{ij}$  distribution has a non-linear behavior. Between the nearest Fe atoms (Fig. 2(a)), depending on the Al concentration, the change is no more than 5 meV, from  $\approx 19$  meV for pure Fe to  $\approx 24$  meV in the

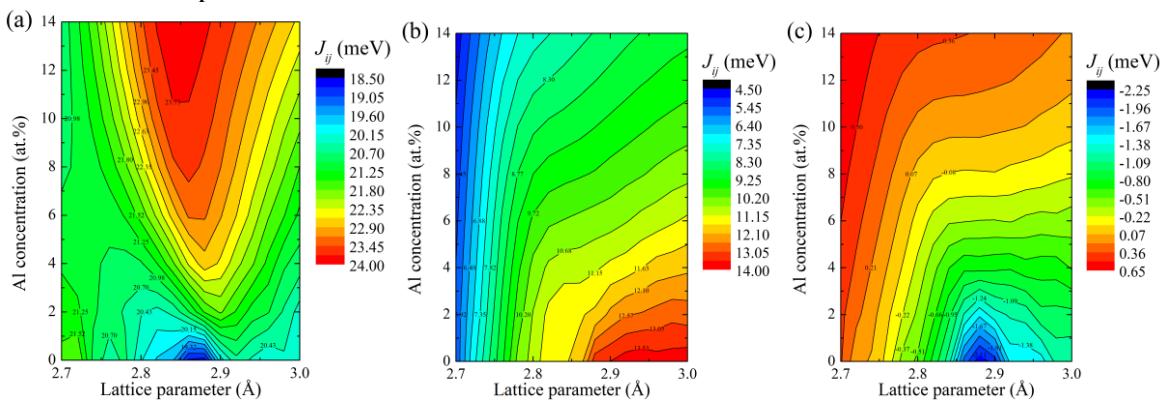


**Fig. 1. Magnetic exchange interaction constants  $J_{ij}$  as a function of the distance ( $d/a$ ) between the pairs of  $i$  and  $j$  atoms, calculated for the experimental lattice value. ( $a_0 = 2,87 \text{ \AA}$ ) [11] in the A2 structure of the  $\text{Fe}_{98}\text{Ga}_2$  alloy**

case of  $\text{Fe}_{86}\text{Al}_{14}$ . Note that such an increase corresponds to the region of lattice parameters  $a = 2,85\text{--}2,86 \text{\AA}$ , which is close to the experimental value. In the range of values  $a \leq 2,76 \text{\AA}$ ,  $J_{ij}$  demonstrate a weak dependence on the Al concentration, changing by 1 meV ( $\approx 20,5 \text{--} 21,5 \text{ meV}$ ), while in the range of  $a \geq 2,92 \text{\AA}$ , the change in  $J_{ij}$  is greater than 3 meV ( $\approx 20 \text{--} \approx 23,5 \text{ meV}$ ).

In the second coordination sphere (at a distance of  $a$ , Fig. 2(b)) the distribution of exchange parameters is different, and two regions can be distinguished in the distribution depending on the lattice parameter. In the first region with small values of the lattice parameters ( $a \leq 2,82 \text{\AA}$ ), the  $J_{ij}$  parameters change slightly with increasing Al concentration, increasing almost linearly with an increase in the lattice parameter at a fixed Al concentration in the alloy. In this region, the parameter values change from  $\approx 4,5$  to  $\approx 10,5 \text{ meV}$ . The second region in Fig. 2(b) is characterized by a decrease in magnetic exchange with an increase in the Al concentration at constant values of  $a$  and a fixed Al concentration in the alloy. In this region, the change in the parameters of the magnetic exchange interaction is observed from  $\approx 8,0$  to  $\approx 14,0 \text{ meV}$ .

In the case of interaction between Fe atoms located at a distance of  $\sqrt{2}$  (Fig. 2(c), the third coordination sphere), a region of antiferromagnetic interaction is observed in the distribution. This region is observed for lattice parameters of  $2,8 \leq a \leq 3,0 \text{\AA}$  and in the concentration range of  $0 \leq x \leq 8 \text{ at. \%}$ . In general, a weak (no more than 4 meV) change in the constants of magnetic exchange interaction in the third coordination sphere can be noted.



**Fig. 2. Distribution of the magnetic exchange interaction constant of  $\text{Fe}_{100-x}\text{Al}_x$  alloys between Fe atoms depending on the crystal lattice parameter  $a$  and the Al concentration  $x$  in (a) the first, (b) second, and (c) third coordination spheres**

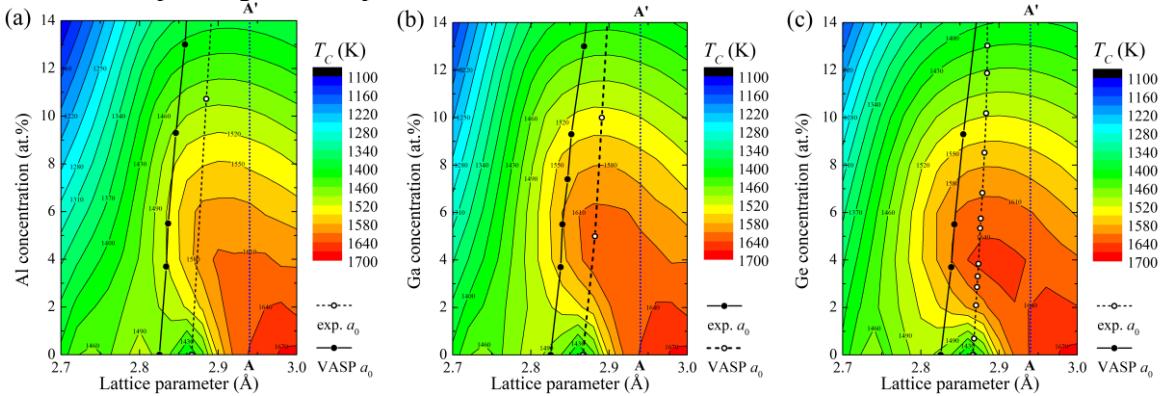
In Fe-Ga and Fe-Ge alloys, the behavior of the magnetic exchange interaction constants is similar. The difference lies in the magnitude of the parameters associated with the presence of  $\text{Me} = \text{Al}, \text{Ga}, \text{Ge}$  atoms in the alloys, which was shown earlier in our works [5–7, 13–17].

In [5–7, 13–17], the authors showed that when modeling using the Monte Carlo method, the calculated Curie temperature values depend significantly on  $J_{ij}$ 's value. Taking this fact into account, in the present work, we analyzed the influence of the value of the crystal lattice parameter and the concentration of nonmagnetic atoms in the  $\text{Fe}_{100-x}\text{Al}_x$ ,  $\text{Fe}_{100-x}\text{Ga}_x$  and  $\text{Fe}_{100-x}\text{Ge}_x$  alloys on the Curie temperature. The values of  $T_C$  shown in Fig. 3 as a function of the crystal lattice parameter and the concentration of  $\text{Me} = \text{Al}, \text{Ga}, \text{Ge}$  atoms were obtained in the mean field approximation, which, as shown by the results of [4, 15], gives overestimated values compared to experimental values, but allows for a qualitative repetition of the experimentally observed trends [4, 18, 19].

Fig. 3 demonstrates that for all three systems: Fe-Al, Fe-Ga, and Fe-Ge, the  $T_C$  distributions are similar, and the main difference is observed in the range of lattice parameters from 2,9 to 3,0  $\text{\AA}$ . Up to lattice parameter values of  $a \geq 2,94 \text{\AA}$ , Curie temperatures have a maximum in the range of  $\text{Me} = \text{Al}, \text{Ga}, \text{Ge}$  atomic concentrations from 0 to 2 at. %. This behavior does not coincide with the observed experimental concentration dependence of the Curie temperature, according to which the behavior of the  $T_C(x)$  curve has a descending behavior when  $\text{Me} = \text{Al}, \text{Ga}, \text{Ge}$  atoms are added to pure iron [12, 20–22]. Note that in the case of the Fe-Ge alloy, a concentration of  $\approx 4,5 \text{ at.\%}$  can be distinguished, where the second maximum is observed at  $a = 2,9 \text{\AA}$ .

Fig. 3 shows the  $T_C(x)$  cross-sections corresponding to the experimental values of the crystal lattice parameter [12, 20, 21] and the crystal lattice parameters obtained by *ab initio* geometric optimization of the crystal structure [6, 7, 23]. For visual clarity, Fig. 4 shows the Curie temperatures as functions of the concentrations of  $\text{Me} = \text{Al}, \text{Ga}, \text{Ge}$  atoms along the experimental and theoretical dependences of the lat-

tice parameters estimated for all the alloys under consideration (Fig. 3). It can be seen that considering both the experimental and theoretical values of  $a_0$  when calculating the  $T_C$  does not allow us to obtain a dependence corresponding to the experimental  $T_C(x)$ .



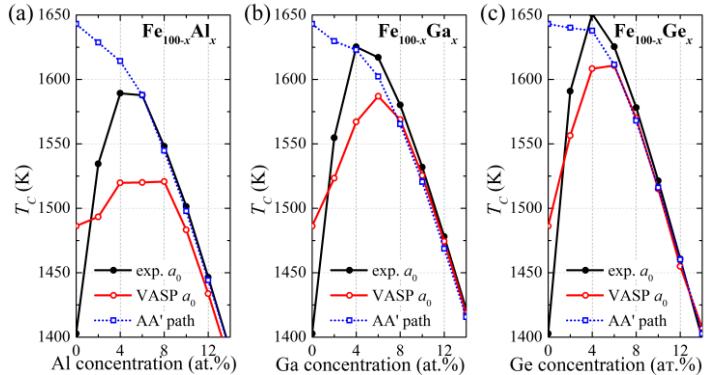
**Fig. 3. Distribution of the Curie temperature for alloys: (a)  $\text{Fe}_{100-x}\text{Al}_x$ , (b)  $\text{Fe}_{100-x}\text{Ga}_x$ , and (c)  $\text{Fe}_{100-x}\text{Ge}_x$ , depending on the crystal lattice parameter  $a$  and the  $Me = \text{Al}, \text{Ga}, \text{Ge}$  concentration  $x$ . Solid (open) circles are experimental (obtained from geometric optimization using the VASP software package) dependences of the crystal lattice parameters; AA' is the distribution cross-section**

The figure shows that maximum Curie temperature corresponds to the concentration range of  $4 \leq x \leq 8$  at. %. The Curie temperature increases below this range and decreases above it. Nevertheless, such a cross-section (AA') can be found in these distributions (Fig. 3), along which the Curie temperature dependence changes from higher to lower with increasing concentration of the alloying element and is qualitatively in agreement with the experimental dependence. This cross-section can be constructed for the parameter  $a \geq 2.94 \text{ \AA}$  for all alloys. As can be seen from Fig. 4, for this value of the crystal lattice parameter, the Curie temperature decreases with increasing concentration of  $Me = \text{Al}, \text{Ga}, \text{Ge}$ .

The final thing to note from the analysis of the Curie temperature distribution is the absence of a direct correlation between the distribution of the largest interactions (between the nearest neighboring Fe atoms) and the distribution of the Curie temperature. From this, we conclude that the value of the Curie temperature is in a complex nonlinear dependence on the magnetic exchange interaction parameters.

## Conclusion

In this work, for  $\text{Fe}_{100-x}Me_x$  alloys ( $Me = \text{Al}, \text{Ga}, \text{Ge}, 0 \leq x \leq 14$  at. %), the constants of magnetic exchange interaction  $J_{ij}$  are calculated within the density functional theory and the Curie temperatures  $T_C$  are estimated within the mean field approximation for the disordered phase A2 in the crystal lattice parameter range of  $2.7 \leq a \leq 3.0 \text{ \AA}$ . The distribution of constants  $J_{ij}$  obtained in the work has a nonlinear behavior. The strongest interaction observed between the nearby Fe atoms is ferromagnetic and is  $\approx 23$  meV. In the second coordination sphere, the ferromagnetic interaction between Fe atoms is preserved and varies within 4–14 meV. In the third coordination sphere, antiferromagnetic exchange interaction is observed, which reaches a value of  $\approx -2.5$  meV. Estimation of the Curie temperature using the mean field approximation yields overestimated values compared to the experimental results. The Curie temperatures obtained with the experimental parameters of the crystal lattice have concentration dependences different from those observed experimentally, namely, they have a maximum Curie temperature in the concentration range of  $4 \leq x \leq 8$  at. %. On the obtained distributions of the Curie temperature depending on the lattice constant and the content of  $Me = \text{Al}, \text{Ga}, \text{Ge}$  in  $\text{Fe}_{100-x}Me_x$  alloys, it is possible to obtain concentration dependences that qualitatively describe the decreasing behavior of  $T_C(x)$  ob-



**Fig. 4. Calculated Curie temperature values for alloys: (a)  $\text{Fe}_{100-x}\text{Al}_x$ , (b)  $\text{Fe}_{100-x}\text{Ga}_x$ , and (c)  $\text{Fe}_{100-x}\text{Ge}_x$ , obtained with experimental, theoretical (by ab initio geometric optimization) parameters of the crystal lattice, and along the cross-section AA' in Fig. 3 ( $a = 2.94 \text{ \AA}$ )**

served experimentally. The study shows that the Curie temperature value is in a complex nonlinear dependence on both the crystal structure constant and the parameters of the magnetic exchange interaction.

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### Information about the authors

Zagrebin Mikhail Aleksandrovich is Dr. Sc. (Physics and Mathematics), Associate Professor, Professor, Radiophysics and Electronics Department, Chelyabinsk State University, Chelyabinsk, Russian Federation, e-mail: miczag@mail.ru.

Markovich Ivan Alekseevich is Post-graduate Student, Mathematical and Computer Modeling Department, South Ural State University, Chelyabinsk, Russian Federation.

Kuznetsov Andrey Sergeevich is Laboratory Assistant, Researcher, Condensed Matter Physics Department, Chelyabinsk State University, Chelyabinsk, Russian Federation.

Matyunina Mariya Viktorovna, is Cand. Sc. (Physics and Mathematics), Associate Professor, Condensed Matter Physics Department, Chelyabinsk State University, Chelyabinsk, Russian Federation, e-mail: matunins.fam@mail.ru.

Butakov Anatoliy Vladimirovich is Cand. Sc. (Physics and Mathematics), Associate Professor, Radiophysics and Electronics Department, Chelyabinsk State University, Chelyabinsk, Russian Federation.

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## КОМПЬЮТЕРНОЕ МОДЕЛИРОВАНИЕ МАГНИТНЫХ СВОЙСТВ НЕУПОРЯДОЧЕННОЙ ОЦК ФАЗЫ СПЛАВОВ $Fe_{100-x}Me_x$ ( $Me = AL, GA, GE$ )

М.А. Загребин<sup>1</sup>, И.А. Маркович<sup>2</sup>, А.С. Кузнецов<sup>1</sup>, М.В. Матюнина<sup>1</sup>, А.В. Бутаков<sup>1</sup>

<sup>1</sup> Челябинский государственный университет, г. Челябинск, Российская Федерация

<sup>2</sup> Южно-Уральский государственный университет, г. Челябинск, Российская Федерация

E-mail: miczag@mail.ru

**Аннотация.** Для сплавов Fe-Me ( $Me = Al, Ga, Ge$ ) в рамках теории функционала плотности рассчитаны постоянные магнитного обменного взаимодействия  $J_{ij}$  и проведена оценка темпера-

туры Кюри  $T_C$  для неупорядоченной фазы A2 в интервале концентраций  $Me = Al, Ga, Ge$   $0 \leq x \leq 14$  ат. % и диапазоне параметров кристаллической решетки  $2,7 \leq a \leq 3,0$  Å. Полученное в работе распределение  $J_{ij}$  демонстрирует нелинейное поведение. Наибольшее ферромагнитное взаимодействие в первой координационной сфере составляет  $\approx 23$  мэВ. В третьей координационной сфере наблюдается антиферромагнитное обменное взаимодействие, которое достигает значения  $\approx -2,5$  мэВ. Температуры магнитного перехода, рассчитанные с использованием приближения молекулярного поля на основании экспериментальных значений и параметров  $a_0$ , рассчитанных в рамках теории функционала плотности, показывают наличие концентрационного интервала  $4 \leq x \leq 8$  ат. %, в котором  $T_C$  возрастает. Сечение распределения рассчитанных значений температуры Кюри показывает, что для всех исследуемых сплавов Fe-Me ( $Me = Al, Ga, Ge$ ) при параметре решетки  $a_0 = 2,94$  Å возможно построить зависимость  $T_C(x)$ , удовлетворяющую экспериментальной.

*Ключевые слова:* сплавы на основе Fe; первопринципные расчеты; магнитное обменное взаимодействие, температура Кюри.

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### Сведения об авторах

Загребин Михаил Александрович – доктор физико-математических наук, доцент, профессор кафедры радиофизики и электроники, Челябинский государственный университет, г. Челябинск, Российская Федерация, e-mail: miczag@mail.ru.

Маркович Иван Алексеевич – аспирант, кафедра «Математическое и компьютерное моделирование», Южно-Уральский государственный университет (национальный исследовательский университет), г. Челябинск, Российская Федерация.

Кузнецов Андрей Сергеевич – лаборант-исследователь кафедры физики конденсированного состояния, Челябинский государственный университет, г. Челябинск, Российская Федерация.

Матюнина Мария Викторовна – кандидат физико-математических наук, доцент кафедры физики конденсированного состояния, Челябинский государственный университет, г. Челябинск, Российская Федерация, e-mail: matunins.fam@mail.ru.

Бутаков Анатолий Владимирович – кандидат физико-математических наук, доцент кафедры радиофизики и электроники, Челябинский государственный университет, г. Челябинск, Российская Федерация.